

Constitution of Calphad multicomponent databases

Nathalie Dupin

Calcul Thermodynamique

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Calphad assessments

- ▶ binary or ternary systems
- ▶ widely available in publications
- ▶ full description based on
 - experimental information
 - ab initio results
- ▶ not always compatible between them
- ▶ basis of multiconstituent databases

Multiconstituent databases

- ▶ many constituents, many phases
- ▶ commercial databases or industrial property
- ▶ reduced composition area of validity
- ▶ simulation of industrial alloys/processes
- ▶ development of new grades

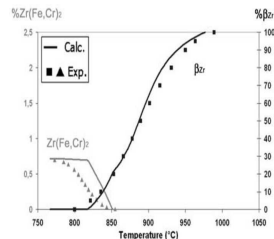
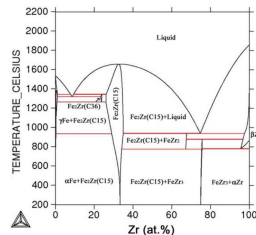
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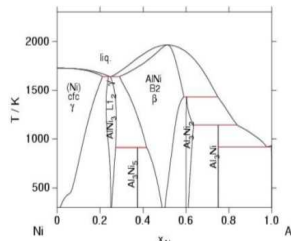
C. Toffolon-Masclat *et al.*
JAI101122



Calphad assessments

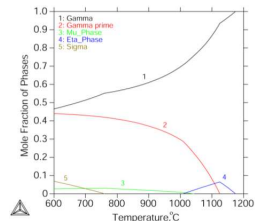
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J. Bratberg et al.
Superalloys 2012



Multiconstituent databases

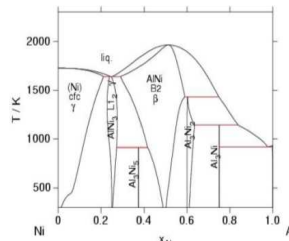
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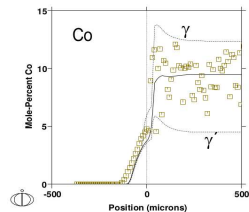
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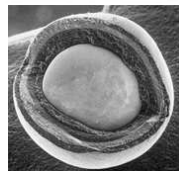
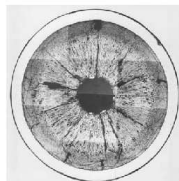
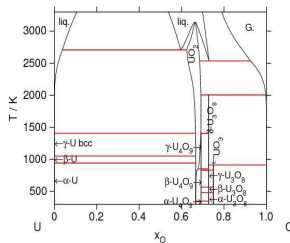


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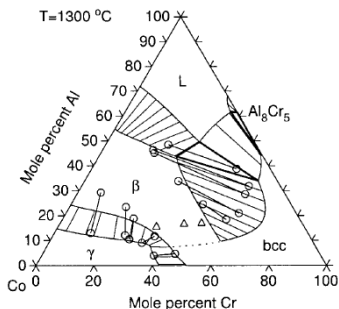
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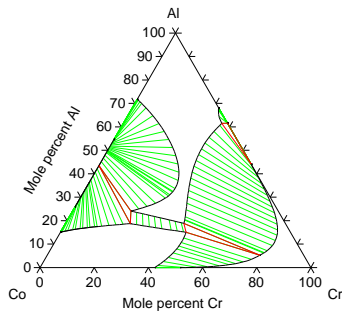


A multicomponent database is not a collection of binaries

T. Gómez-Acebo et al. JPEDAV (2004) 25:237-251

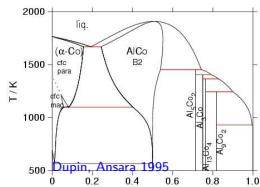


exp.: Ishikawa et al. *Ber.Bun/Phys.Chem.* **102** 1206 (1998)

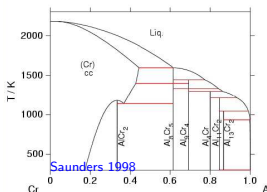


TCNI β and bcc : (Al,Co,Cr,Va)(Al,Co,Cr,Va)

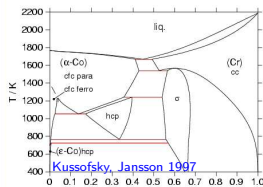
β : (Al,Co)(Co,Va)



Dupin, Ansara 1995



Saunders 1998




Kussovsky, Jansson 1997


Constitution of a multicomponent database

- ▶ Definition of the elements and phases of interest
- ▶ Choice of the models
 - Modification of existing assessments
- ▶ Assessment of missing constituting systems
- ▶ Metastable inputs based on ternary knowledge/guess
- ▶ Validation


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...

- ▶ Feedback of users
- ▶ New project
 - revision of some phases
 - wider validity range
 - more elements

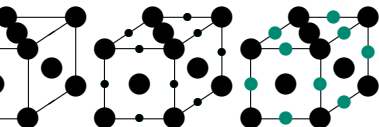
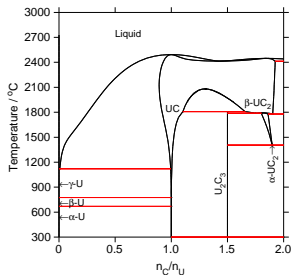
Choice of models

- ▶ Same crystallographic structure \Rightarrow same model
 - not always that obvious
- ▶ Crystallographic analysis allows description of
 - ordering relationship
 - thermodynamics
 - phase diagram
- ▶ Good models “work” in all systems.
- ▶ FP results help.

Defects: phase diagram and thermodynamics

(U)(C, \square ,C₂)

C. Guéneau *et al.* JNM 419 (2011) 145-167



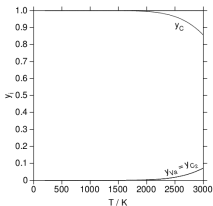
A1

B1

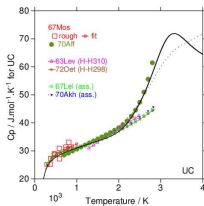
Cu

NaCl

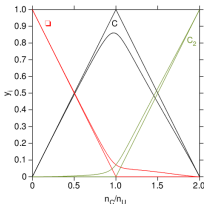
KCN



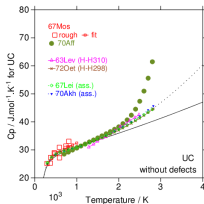
vs temperature for $x_C=0.5$



(U)(C,C₂, \square) at $x_C=0.5$

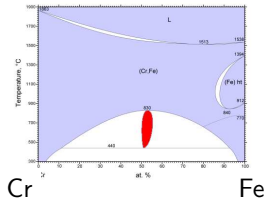
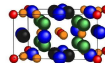


vs composition at 1500 and 3000K

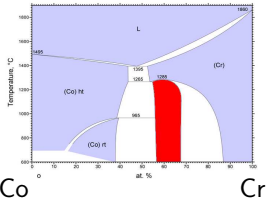


(U)(C)

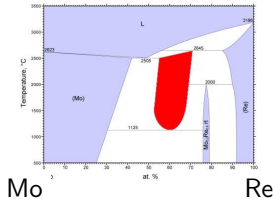
Good models



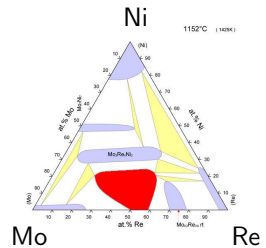
Cr Fe



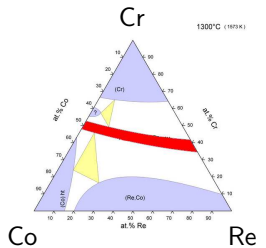
Co Cr



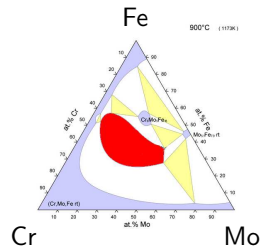
Mo Re



Mo Re



Co Re

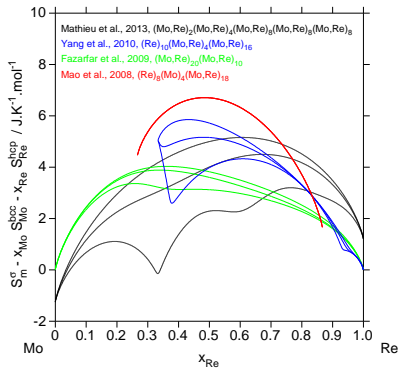
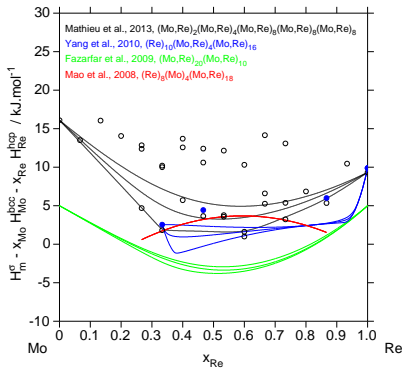
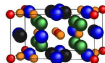


Cr Mo

The σ phase presents different ordering degrees, different topology.
A good model should describe this in all the systems considered.

Good models

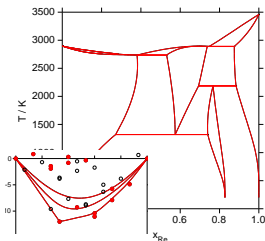
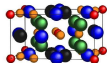
R. Mathieu et al. Calphad 43 (2013) 18-31



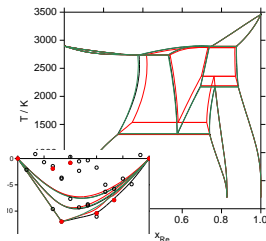
A good model should be able to separate $S^{conf.}$ from $S^{vib.}$.

Good models

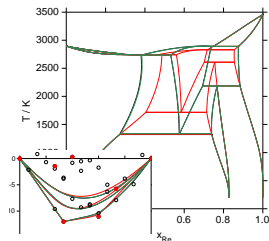
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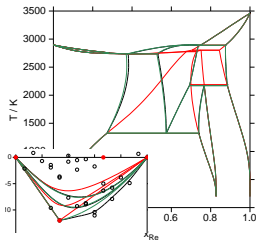
→4SL: $(A,B)_{10}(A,B)_4(A,B)_8(A,B)_8$
 $L(\text{SIGMA}, \text{MO}, \text{RE}: *, *, *, 0) = 0$



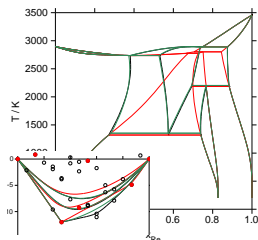
→3SL-23: $(A,B)_{10}(A,B)_{12}(A,B)_8$
 $L(\text{SIGMA}, \text{MO}, \text{RE}: *, *, *, 0) = 0$
 $L(\text{SIGMA}, *, \text{MO}, \text{RE}: *, 0) = -45800 + 7.8 * T$



→3SL-25: $(A,B)_{10}(A,B)_{12}(A,B)_8$
 $L(\text{SIGMA}, \text{MO}, \text{RE}: *, *, *, 0) = 0$
 $L(\text{SIGMA}, *, \text{MO}, \text{RE}: *, 0) = -67750 + 2.8 * T$



→2SL: $(A,B)_{10}(A,B)_{20}$
 $L(\text{SIGMA}, \text{MO}, \text{RE}: *, *, *, 0) = 0$
 $L(\text{SIGMA}, *, \text{MO}, \text{RE}: 0) = -242810 + 11.4 * T$

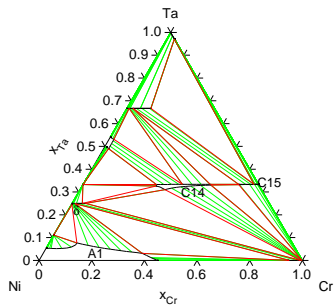


→3SL-35: $(A,B)_{10}(A,B)_4(A,B)_{16}$
 $L(\text{SIGMA}, \text{MO}, \text{RE}: *, *, *, 0) = 0$
 $L(\text{SIGMA}, *, *, \text{MO}, \text{RE}: 0) = -130000 - 1.1 * T$

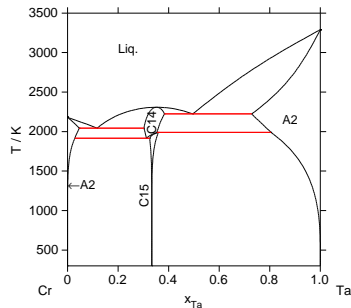
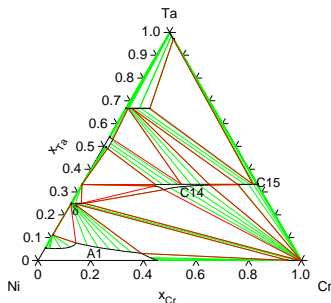
A good model should be able to separate H and S contributions.

Recommended 3SL :
 $(A,B)_{10}(A,B)_{12}(A,B)_8$
 rather than classical
 $(A,B)_{10}(A,B)_4(A,B)_{16}$

Assessment of metastable states

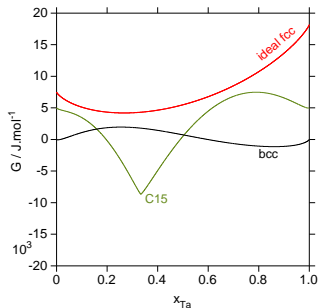
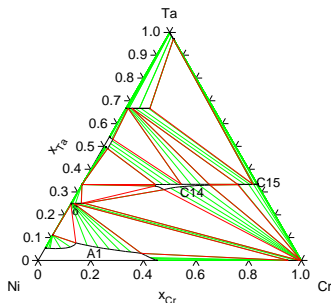


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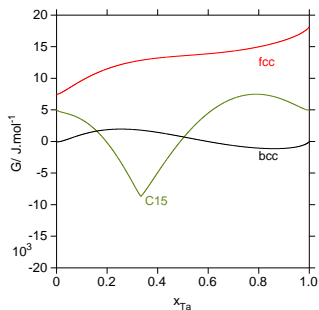
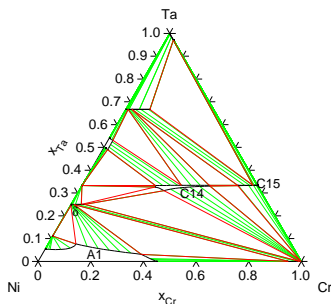
In order to describe the **fcc** phase in the ternary Cr-Ni-Ta, it has to be described in the binary Cr-Ta where it is not stable. It has to be identical whatever system implying Cr-Ta system. Nowadays, FP results help in such assessment.

Assessment of metastable states



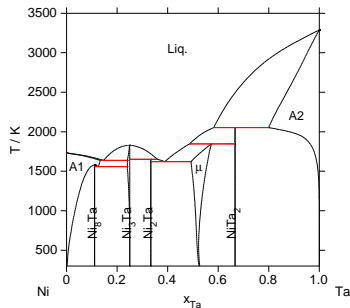
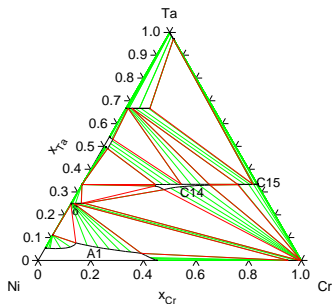
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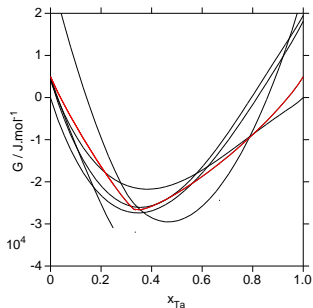
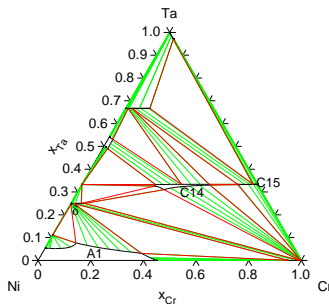
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Assessment of metastable states



In order to describe the **C14** stability in the ternary Cr-Ni-Ta, the binary metastable C14 phase in Ni-Ta has to be described. It has to be identical whatever system implying the Ni-Ta system. Care should be given to keep the phase metastable in the binary.

Assessment of metastable states

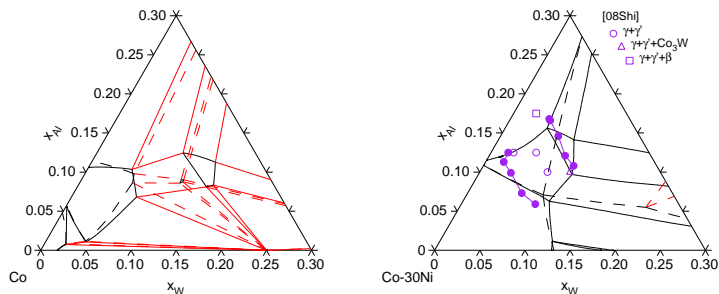


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Always under improvement

The stability of the γ' phase in the ternary Al-Co-W was reported for the first time about 10 years ago. The experimental knowledge of the γ' phase extension from this area has rapidly improved as well as its description.

J. Bratberg *et al.* Superalloys 2012



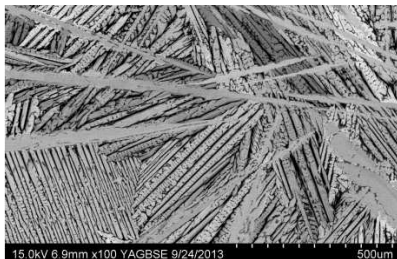
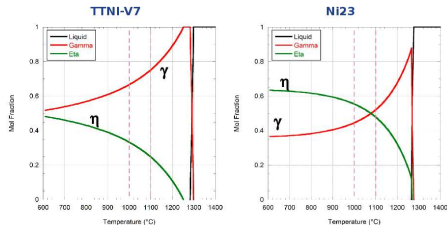
Full lines TCNI5 (2011), dashed lines TCNI4 (2008)

[08Shi] Shinagawa *et al.* *Mat. Tr.* **49** 1474 (2008)

Always under improvement

Ni-5Ta-12.9Ti (at.%)

GE property: Raghav Adharapurapu,
Rich DiDomizio, Ian Spinelli



- ▶ TTNI-V7
 γ primary phase,
 η formed in solid state
- ▶ NI23 (TCNI5)
 γ primary phase,
 η from the liquid

- ▶ experimentally
 η primary phase

- ▶ as cast and
equilibrated states
experimental study
will help a new
assessment

To constitute multicomponent databases

- ▶ Good models are mandatory
- ▶ FP results help
- ▶ Experimental knowledge is key

Thanks

Presented during the TMS annual Meeting: Hume-Rothery Award
Symposium: Multicomponent Alloy Metallurgy, the Bridge from
Materials Science to Materials Engineering,
Orlando, FL, March 16-19, 2015
dedicated to W.J. Boettinger

The view 18 of the present version has been slightly amended since
after a question formulated by Ursula Kattner during the lecture.