

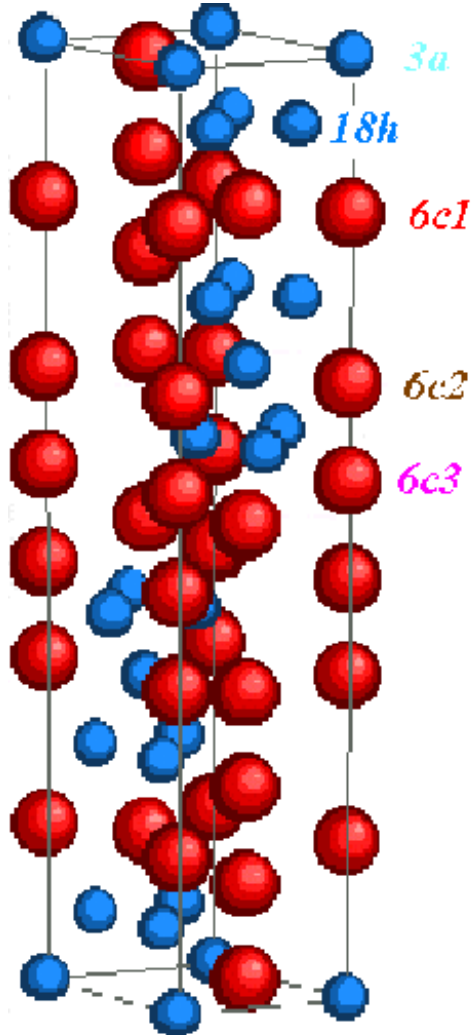
Study of the μ phase of Ni-Nb
Rietveld analysis,
First principle calculations
and Calphad approach

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Outline

- The μ phase
- The present study
- Preliminary results
- Perspectives

The mu phase



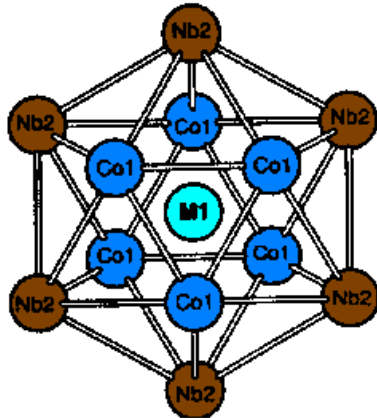
Ideal Structure

Prototype μ	Fe ₇ W ₆				
Space group Pearson symbol	$R\bar{3}m$ $hR13$				
Wyckhoff	3a	6c1	6c2	6c3	18h
Coordination number	12	15	16	14	12
Site occupation	Fe	W	W	W	Fe

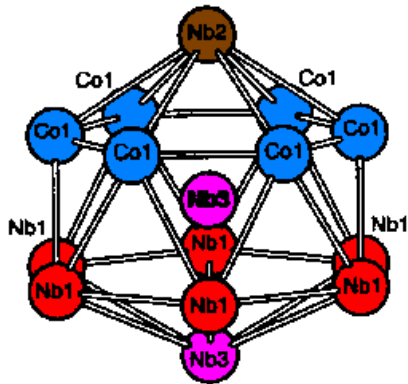
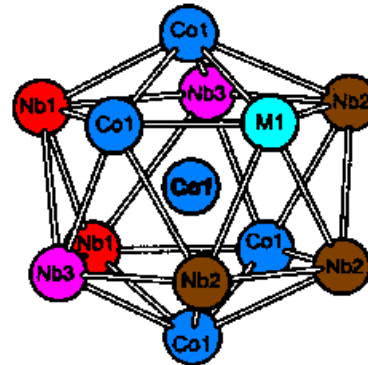
The mu phase

Not stoichiometric

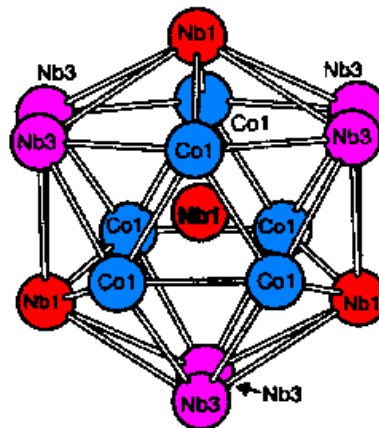
3a
CN 12



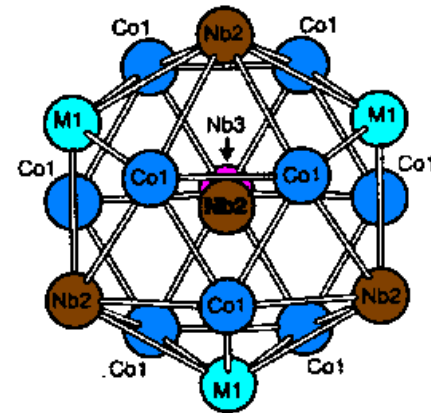
18h
CN 12



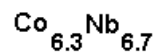
6c3 CN 14



6c1 CN 15



6c2 CN 16



The mu phase

Model ?

Crystallography

1995 Wagner *et al.* (Co_{6.3}Nb_{6.7})

in agreement with

1962 Kripryskaya *et al.* (NiTa and NiNb)

⇒ suitable model for NiNb :

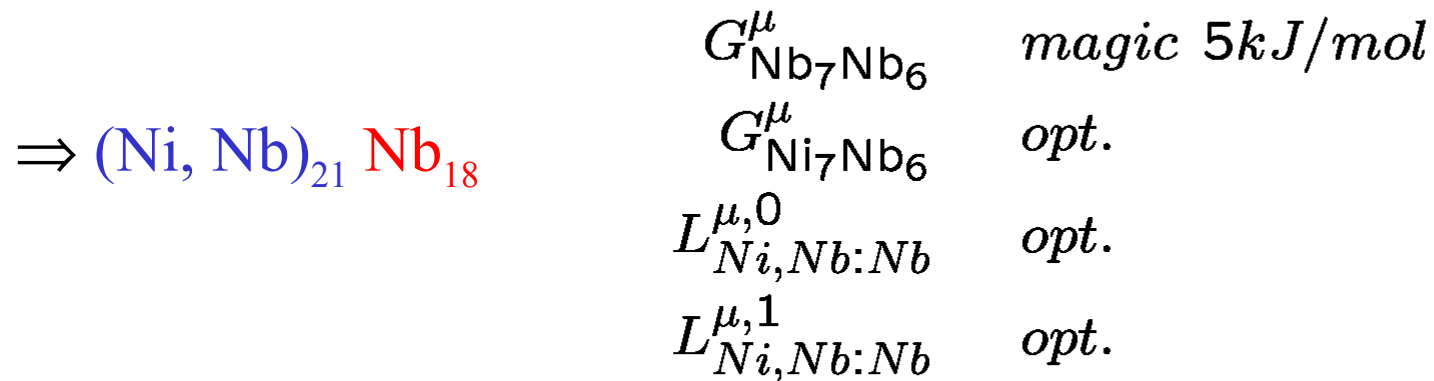


The mu phase

Available

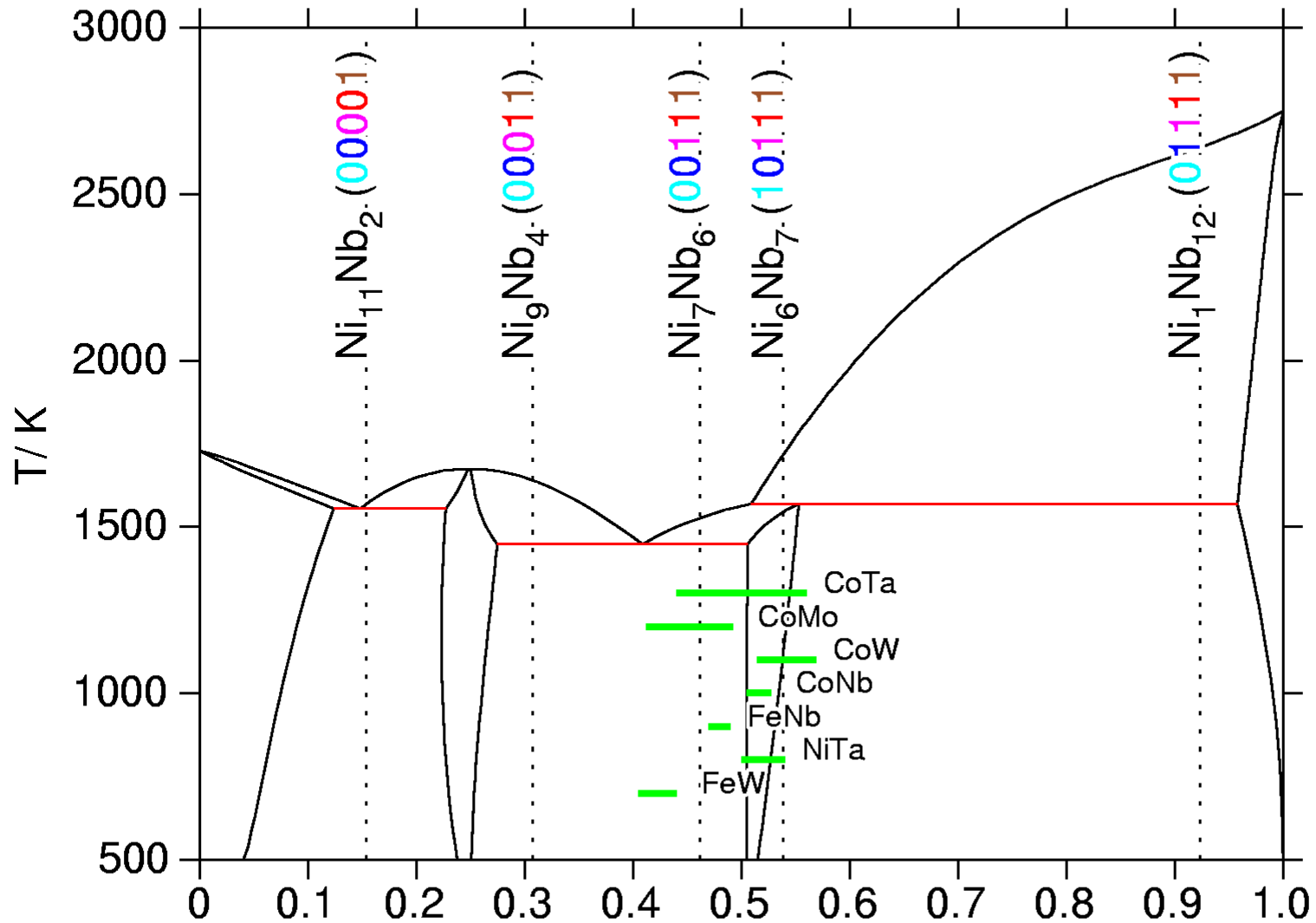
1996 Bolcavage and Kattner

- phase diagram
- CN 12 for 3a and 18h
- compatibility with other descriptions



The mu phase

Composition range

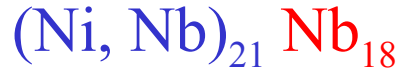


The mu phase

Model?



doesn't allow to describe phase diagram



does describe the phase diagram

doesn't describe the crystallography

cannot describe all the μ phases

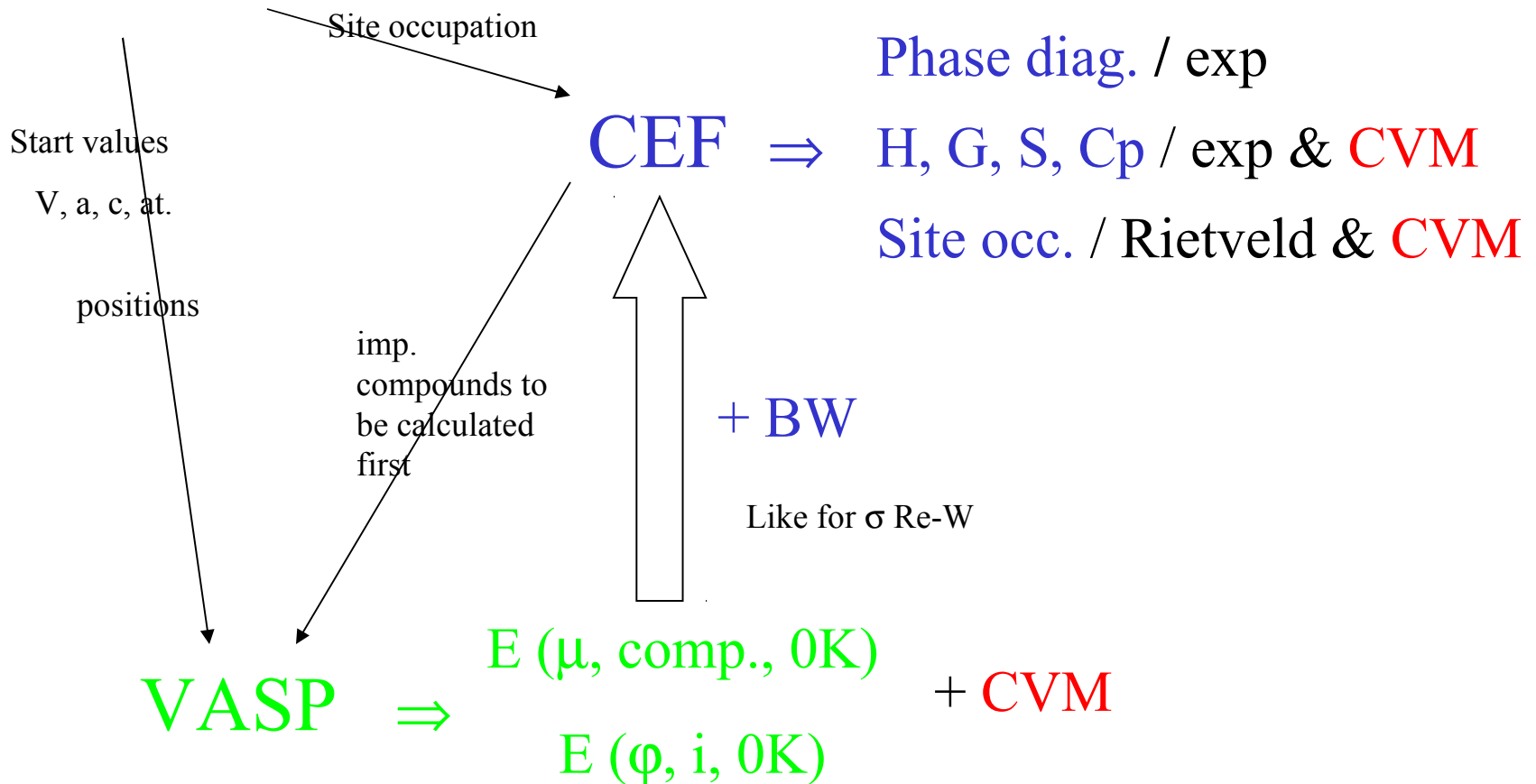
A more complex model is needed.

How to assess easily all its parameters?

The present study

Approach

Rietveld



The present study

VASP

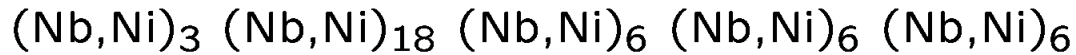
Total Energies computed by Vienna Ab-initio Simulation Package

- **DFA** Density Functional Approximation
- **GGA** Generalized Gradient Approximation
- **non-local optimized ultrasoft pseudopotentials**
- **Full relaxed** (Volume-shape and atomic positions)

For more details: Journal of Alloys and Compounds: Berne, Sluiter, and Pasturel, Theoretical Approach of phase selection in refractory metals and alloys, 334 (2002) 21-33

The present study

C. E. Formalism



$$G_m^\mu(T) - \sum_i x_i^\mu H_i^{\text{SER}}(T_0)$$

$$= \Delta G(x) + \Delta G(y)$$

$$= \sum_i x_i^\mu (G_i^\mu(T) - H_i^{\text{SER}}(T_0)) \Rightarrow \text{Ref}$$

$$+ \sum_C y_i^1 y_j^2 y_k^3 y_l^4 y_m^5 (G_C^\mu(T) - a_{\text{Nb}} G_{\text{Nb}}^\mu(T) - a_{\text{Ni}} G_{\text{Ni}}^\mu(T)) \Rightarrow \text{Ref}$$

$$+ R T \sum_s \alpha_i^s \sum_i y_i^s \ln y_i^s \Rightarrow \text{id}$$

$$+ \sum_\nu x_{\text{Nb}}^\mu x_{\text{Ni}}^\mu (x_{\text{Nb}}^\mu - x_{\text{Ni}}^\mu)^\nu L_{\text{Nb,Ni}}^{\mu,\nu} \Rightarrow \text{xs}$$

x : at. composition

y : site fraction

The present study

Parameters

Ref. from VASP

$$\left(G_i^\mu(T) - H_i^{SER}(T_0) \right) \text{ from VASP}$$

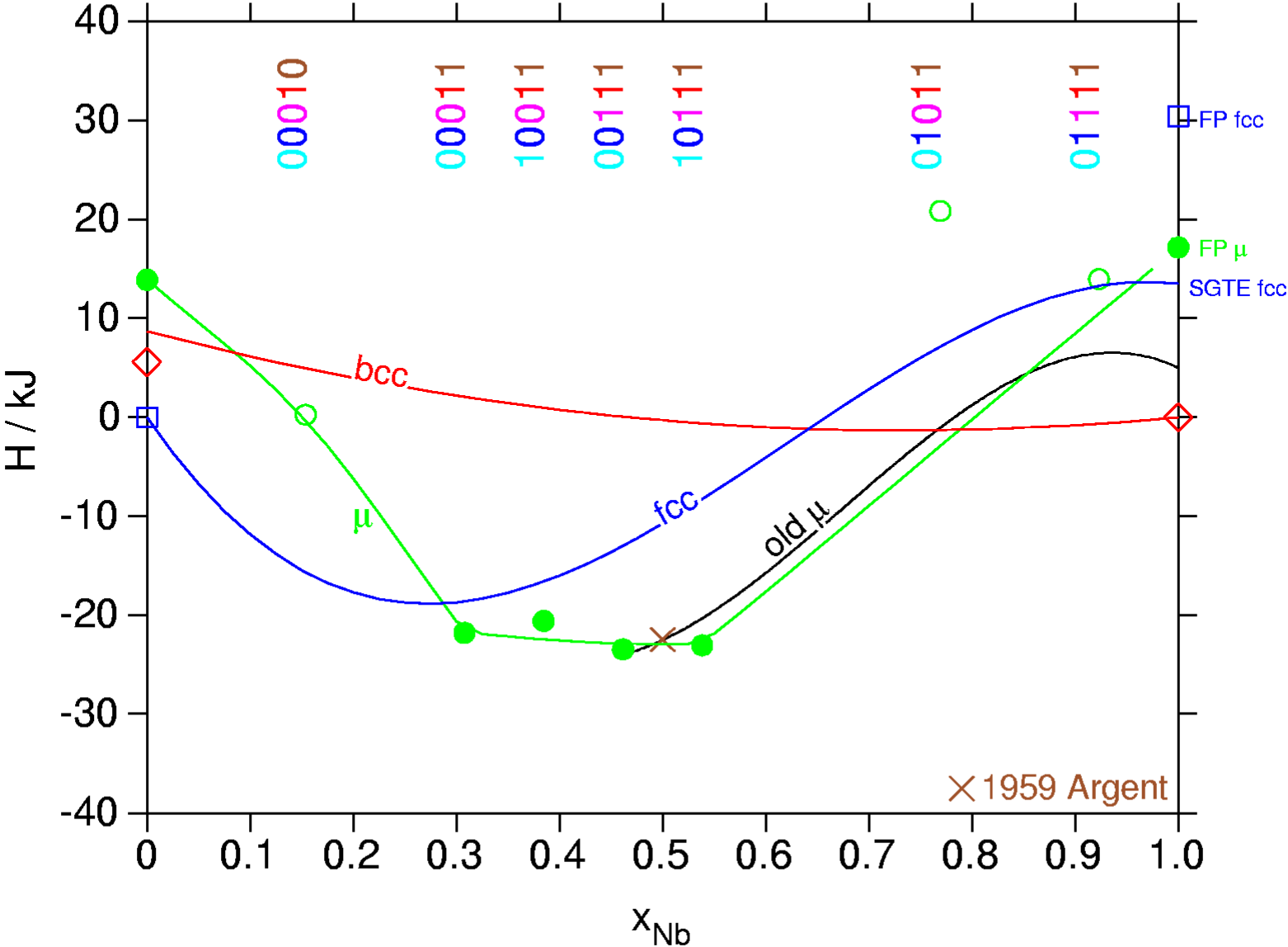
$$\left(G_C^\mu(T) - a_{Nb} G_{Nb}^\mu(T) - a_{Ni} G_{Ni}^\mu(T) \right) \text{ from VASP}$$

11 available over 32

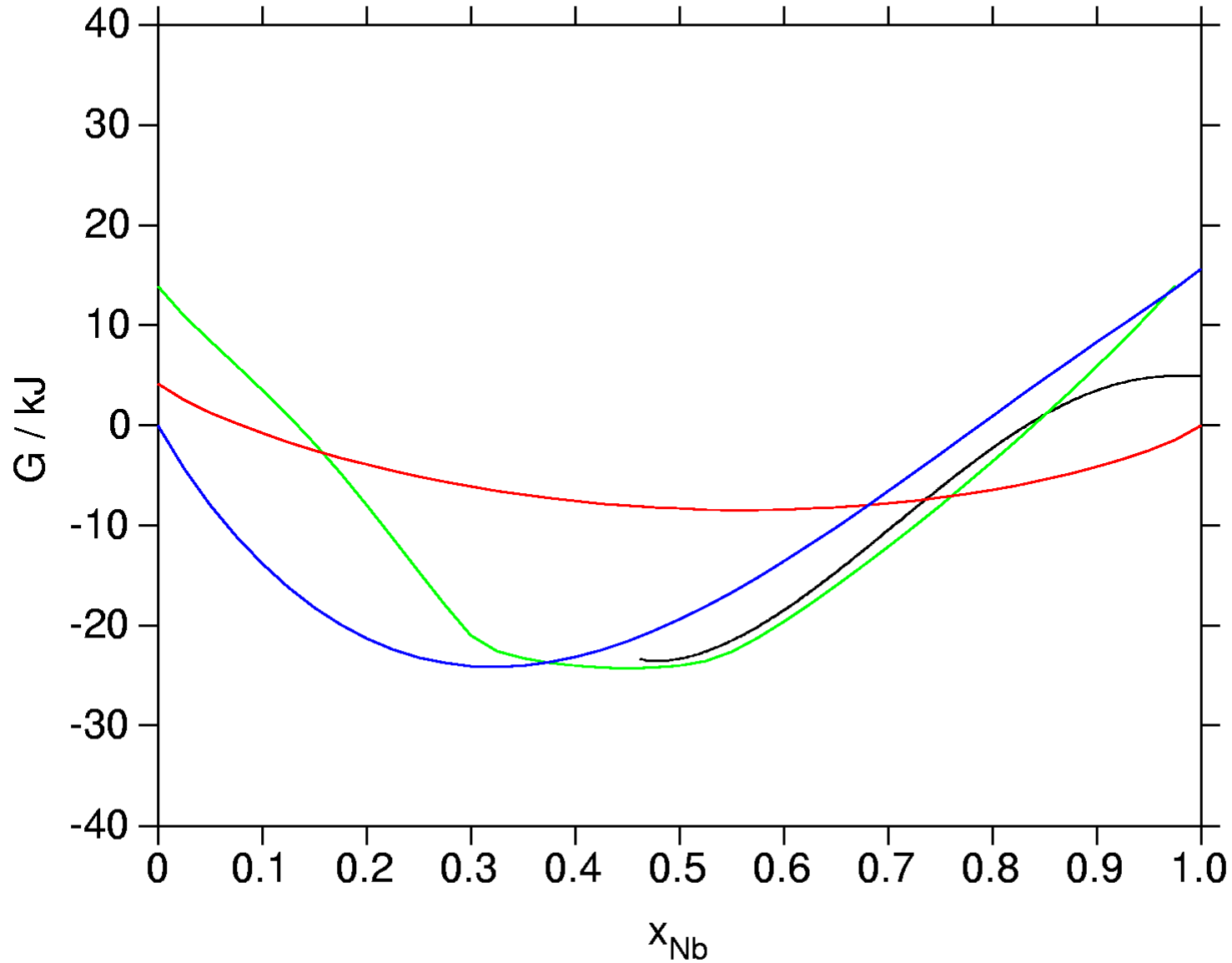
No xs !

$$L_{Nb, Ni}^{\mu, \nu} \text{ zero}$$

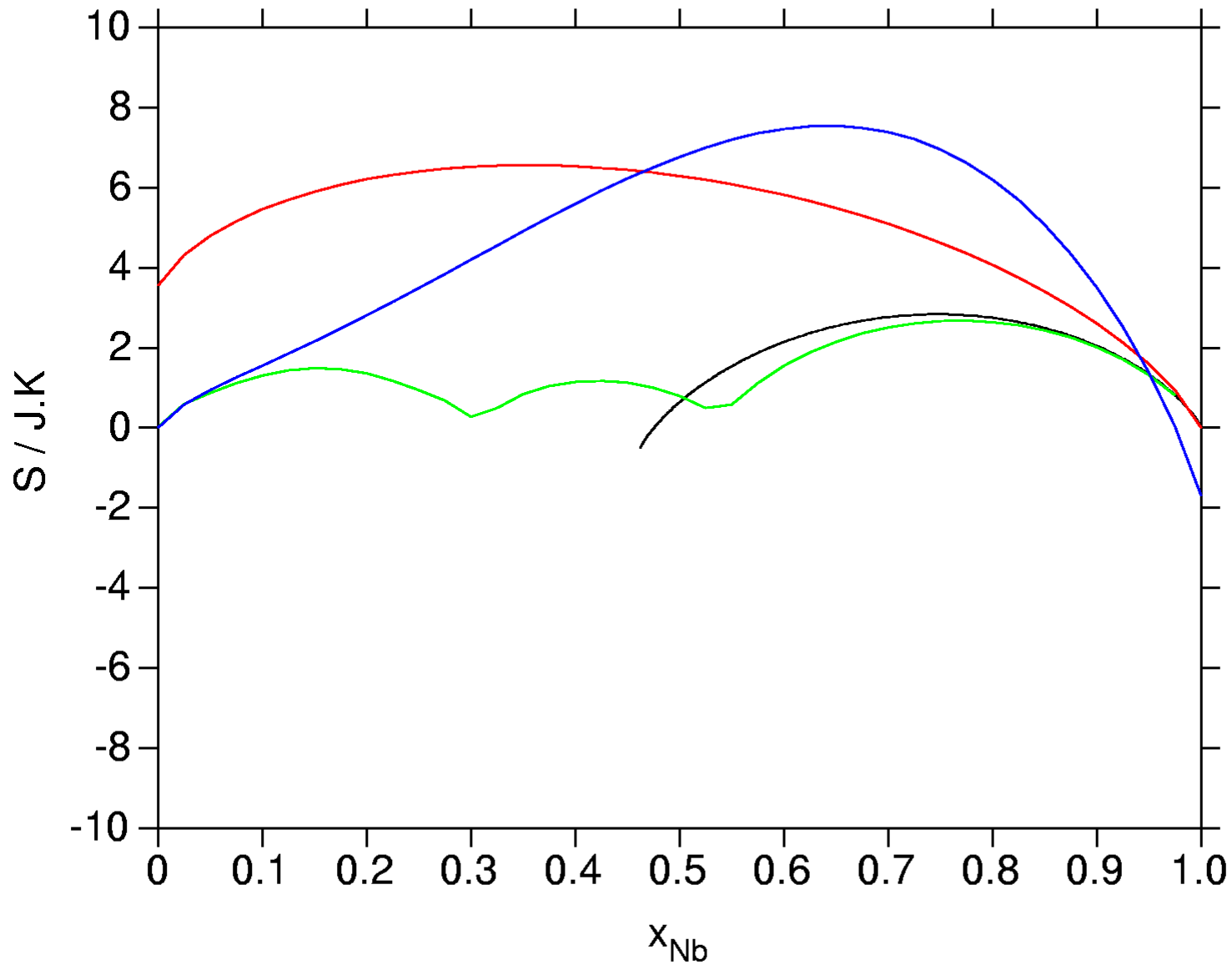
Preliminary results



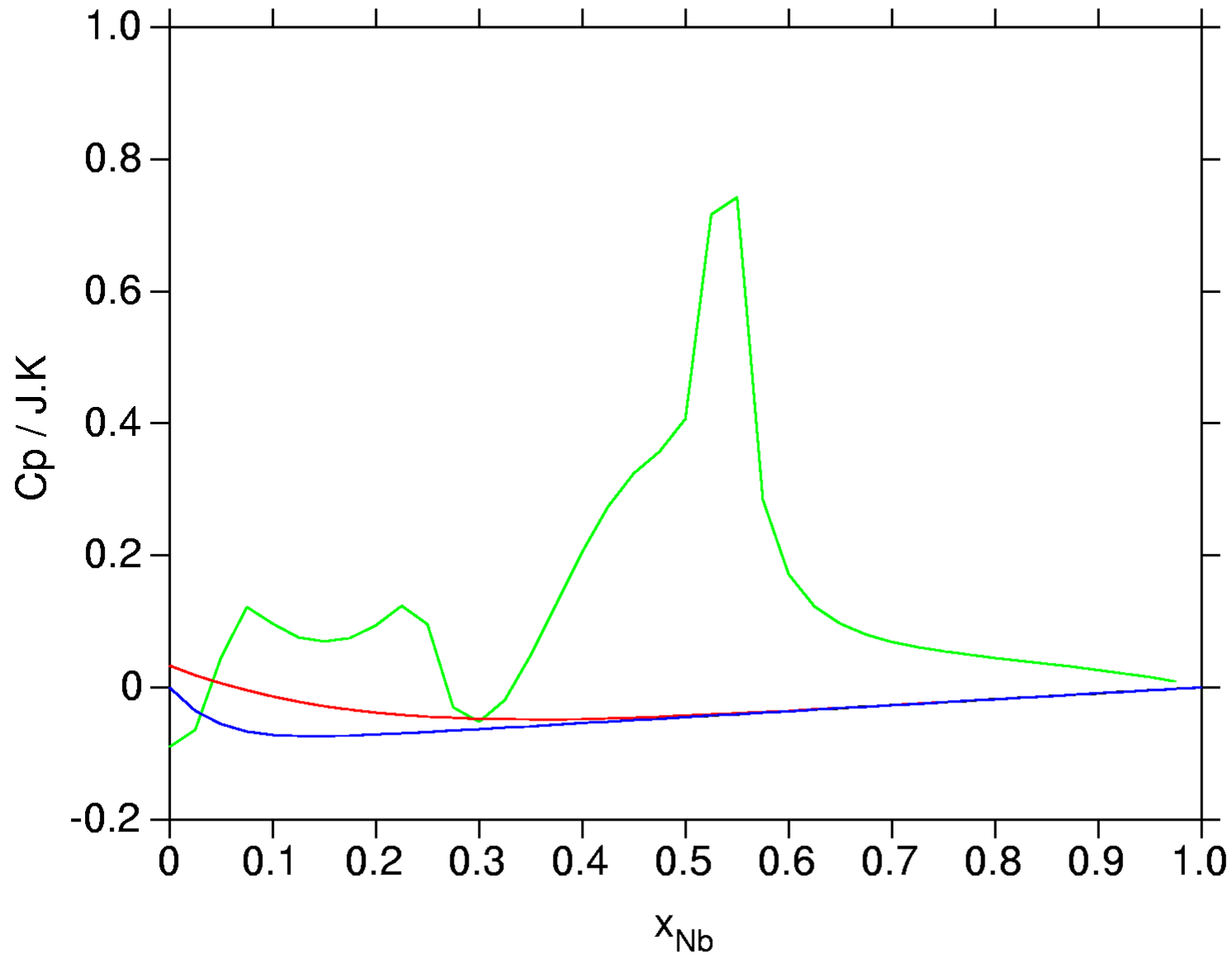
Preliminary results



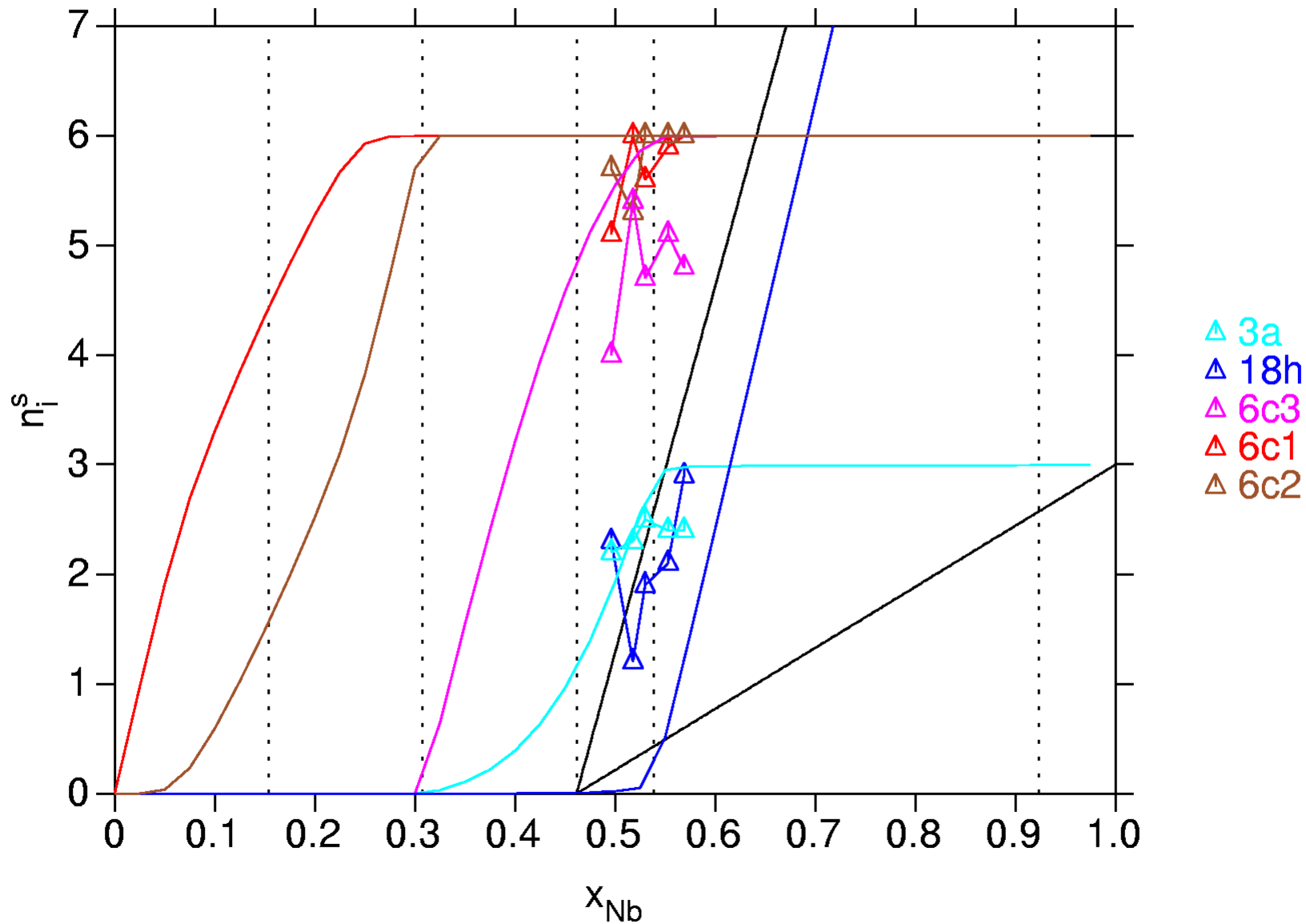
Preliminary results



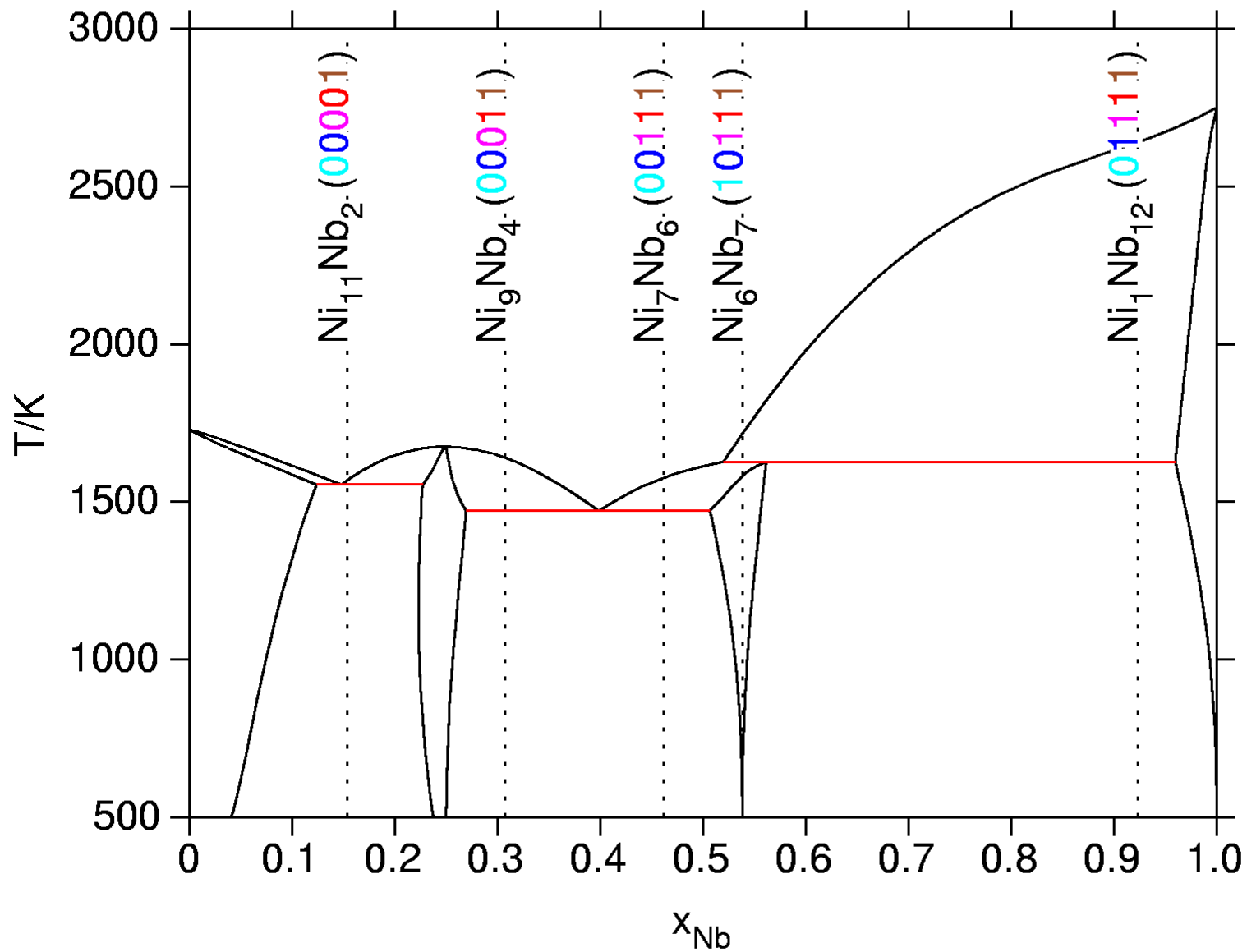
Preliminary results



Preliminary results



Preliminary results



Perspectives

- Preliminary results very promising but
 - missing compounds to be calculated
 - comparison of H, G, S, Cp, y with CVM, ...
- Application of the approach to
 - other systems
 - other phases
- Revision of lattice stabilities?
 - μ , σ , Laves, for pure elements
 - fcc Nb